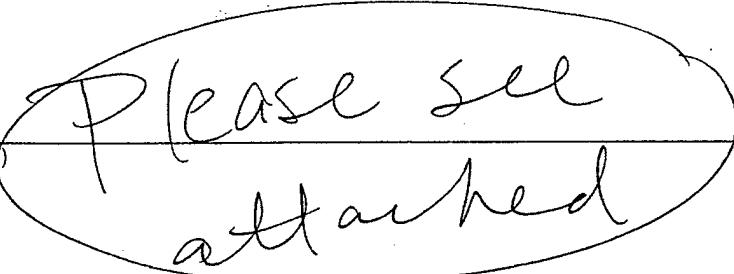


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1. REPORT DATE (DD-MM-YYYY)	2. REPORT TYPE	3. DATES COVERED (From - To)		
	Technical Papers			
4. TITLE AND SUBTITLE			5a. CONTRACT NUMBER	
			5b. GRANT NUMBER	
			5c. PROGRAM ELEMENT NUMBER	
			5d. PROJECT NUMBER	2302
			5e. TASK NUMBER	M162
			5f. WORK UNIT NUMBER	346120
7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES)			8. PERFORMING ORGANIZATION REPORT	
Air Force Research Laboratory (AFMC) AFRL/PRS 5 Pollux Drive Edwards AFB CA 93524-7048				
9. SPONSORING / MONITORING AGENCY NAME(S) AND ADDRESS(ES)			10. SPONSOR/MONITOR'S ACRONYM(S)	
Air Force Research Laboratory (AFMC) AFRL/PRS 5 Pollux Drive Edwards AFB CA 93524-7048			11. SPONSOR/MONITOR'S NUMBER(S)	
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12. DISTRIBUTION / AVAILABILITY STATEMENT				
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13. SUPPLEMENTARY NOTES				
14. ABSTRACT				
				
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				19a. NAME OF RESPONSIBLE PERSON Leilani Richardson
a. REPORT	b. ABSTRACT	c. THIS PAGE		
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2302M1G2 *1999-2001*

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20 Apr 2001

SUBJECT: Authorization for Release of Technical Information, Control Number: **AFRL-PR-ED-TP-2001-100**  
Smith, C.W.; Constantinescu, D.M.; Liu, C.T., "Stress Intensity Factor and Paths for Cracks in  
Photoelastic Motor Grain Models Under Internal Pressure"

**2001 ASME Winter Meeting**  
**(New York, NY, 5-9 Nov 01) (Deadline: 24 Apr 01)**

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# STRESS INTENSITY FACTORS AND PATHS FOR CRACKS IN PHOTOELASTIC MOTOR GRAIN MODELS UNDER INTERNAL PRESSURE

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## ABSTRACT

Computational analysis and two-dimensional tensile tests on single motor grain fins suggest that cracks in fin tips are most likely to originate at the coalescence of a fin end tip radius, with a small radius from the side of the fin. Some manufacturers have also noticed defects formed during casting at the fin tip on the fin axis. The following is an experimental investigation utilizing frozen stress photoelastic models of an existing motor grain geometry in order to clarify stress intensity factor (SIF) values and crack growth paths for cracks emanating from the two above-noted potential critical loci. Comparisons between results from cracks grown from the two loci will be made, suggesting interesting conclusions.

## INTRODUCTION

Two-dimensional thermal shrinkage tests on photoelastic models and two-dimensional tensile tests on single fin sections of motor grain have suggested that, for a specific fin geometry consisting of a small edge fin radius coalescing into a large fin tip radius, the critical locus for crack initiation lies at the point of confluence for the two radii at the fin tip. On the other hand, some motor grain manufacturers have reported

cracks emanating from the fin tip along its own axis of symmetry as a result of voids collecting there during the casting process. One aspect of the problem which has not been explored is how cracks would grow from these two critical points in a three-dimensional model and the corresponding stress intensity factors.

The first author and his colleagues have achieved some success (Smith, 1990), (Smith and Wang, 1992), (Smith, Wang and Liu, 1995), (Smith, Liu and Wang, 1995) combining the frozen stress photoelastic method (Appendix A) together with a two-parameter method (Smith and Kobayashi, 1993) for analyzing local stress fields around crack tips for (SIF) determination in generic and specific rocket motor geometries. The present paper describes an attempt to use this method to study the crack growth and SIF values for cracks emanating from the two critical loci described above in three-dimensional models under internal pressure.

## TEST MODELS AND PROCEDURES

All tests models were cast at Measurements Group (Raleigh, NC) in molds provided by Virginia Tech using PLM-4BR stress freezing material manufactured by Measurements Group. The model dimensions are pictured in Fig. 1. The starter

cracks were made by first drilling a small hole opposite the fin in which the crack was to be located, sliding a shaft with a tip blade into the hole, positioning the blade at the critical point on the fin surface and then striking the shaft with a hammer. The starter cracks then emanated from the blade tip into the material as "natural" or real cracks. For the cracks located at the point of confluence of the small fin edge radius with the larger fin tip surface radius, a shorter tip blade was used opposite the main blade to balance the impact load on the shaft (Fig. 2). This shorter blade caused some slight damage and sometimes a small crack which was patched with PMC-1 cement. Of course, cracks emanating from the critical locus on the axis of symmetry required no balance blade and were more easily produced. After the starter cracks were made, the drilled holes were plugged and sealed.

It is important that crack growth be achieved from these starter cracks under pressure in order to establish the "pressure" crack growth path as opposed to the starter crack generated by wedge forces.

The test setup is pictured in Fig. 3. The pressure was monitored only outside the model, as indicated, and maintained its value when end caps leaked or when a crack grew through the model wall. In addition to the internal pressure, an end load was applied to the model to approximately simulate the effect of an open end as the motor grain may experience in service.

Four identical models have been tested to date. Model #1 was utilized solely to determine the stress fringe order at the two critical loci without cracks (Fig. 4). Since the test material showed a slight deterioration in the surface layer, graphs of fringe order versus path along "squeezed" fringes were extrapolated to the

boundary and showed little difference in fringe value at the two critical points (Fig. 5).

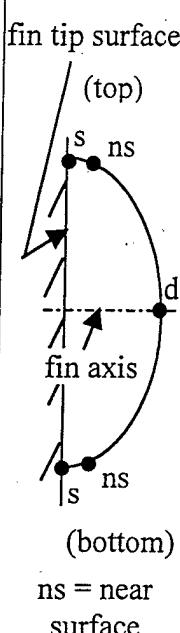
Models #2, #3 and #4 each contained two cracks; in each case in fins adjacent to an uncracked fin and one with a plugged hole in order to allow independent analysis at each crack (Fig. 6). After stress freezing and removal of thin slices mutually orthogonal to the local crack surface and its border, the slices were analyzed by a two-parameter model, briefly described in Appendix B.

## RESULTS

As noted above, two types of cracks were used in the experiments. Cracks which were symmetric with respect to both load and geometry were initiated at the fin tip along the axis of the fin. These cracks were planar, semi-elliptic cracks which maintained these characteristics as they grew. SIF values were determined at maximum depth, along or near to the fin surface, and exhibited pure Mode I SIF values. Two of these cracks penetrated the outer surface of a model and readily grew at smaller load levels than the off-axis cracks which emanated at the locus of confluence of the two fin tip radii. These off-axis cracks were generally non-planar and were extremely difficult to grow even at elevated pressures. This was because the tips of these cracks were under mixed mode load and had to turn a significant amount in order to grow. In fact, with one exception, none of these cracks reached a purely Mode I state during testing.

The test data and results are presented in a normalized form in Table I. Since stress freezing pressure was maintained after breakthrough, data did not appear to be adversely affected. However, fringe observations on Model 1 and SIF data on one end of the long

Table I

Loads <sup>1</sup>	Crack Description <sup>2</sup>	$F_i = K_r \sqrt{\mathcal{Q}} / p_{sf} \sqrt{\pi a^3} \quad i = 1, 2$			
		depth (d)	surface (s)		
			top	bottom	
$P = 311.4 \text{ N}$ $p_{max} = 0.145 \text{ MPa}$ $p_{sf} = 0.07 \text{ MPa}$	<b>Model 2</b>				
	Long off-axis crack $a/c = 0.66 \quad a/t = 0.26$	$F_1 = 2.03$ $F_2 = 0$	2.42	2.90	
	Long symmetric crack $a/c = 0.9 \quad a/t = 1.0$	penetrated	—	2.72	
	<b>Model 3</b>				
	Short symmetric crack $a/c = 0.5 \quad a/t = 0.41$	1.67	2.25	1.82	
	Long symmetric crack $a/c = 0.68 \quad a/t = 1.0$	penetrated	2.00	2.16	
$P = 88.97 \text{ N}$ $p_{max} = 0.049 \text{ MPa}$ $p_{sf} = 0.035 \text{ MPa}$	<b>Model 4</b>				$t = \text{distance from fin tip to outer surface along fin axis, (37.08 mm)}$
	Right off-axis $a/c = 0.81 \quad a/t = 0.25$	$F_1 = 1.51+$ $F_2 = 0.47+$	—	—	
	Left off-axis $a/c = 0.78 \quad a/t = 0.23$	$F_1 = 1.90$ $F_2 = 0.48$	ns	ns	
			2.62	2.55	

+ Opposite crack surfaces were not uniformly separated through the slice and this led to local fringe distortions.

Notations:

- 1 -  $P$  = axial compressive load  
 $p_{max}$  = maximum internal pressure to grow crack  
 $p_{sf}$  = stress freezing pressure
- 2 -  $a$  = crack length  
 $c$  = half length of crack in fin tip surface
- 3 -  $\sqrt{\mathcal{Q}}$  = approximation of elliptic integral of second kind

$$\mathcal{Q} = 1 + 1.464 \left( \frac{a}{c} \right)^{1.65} \quad \frac{a}{c} \leq 1$$

$$\mathcal{Q} = 1 + 1.464 \left( \frac{c}{a} \right)^{1.65} \quad \frac{a}{c} > 1$$

All flaws were characterized as semi-elliptic flaws of depth  $a$  and length  $2c$ . However, off-axis cracks were neither perfectly semi-elliptic nor planar.

symmetric crack in Model 2, together with surface values for the right off-axis crack in Model 4, suggested a small skin effect in the material, so these values were omitted. In order to escape this effect, values of  $F_1$  were obtained 20° off of the surface or near the fin surface "ns" in Model 4 and appeared quite reasonable. All of the cracks grew more along the axis of the model than through the thickness.

The long off-axis crack in Model 2 was the only such crack to grow sufficiently to escape the shear mode. It is pictured in Fig. 7.

### SUMMARY

Based on the above studies to date, it has been found that:

- i. Cracks emanating from a fin tip on the axis of the fin exhibit pure Mode I and grow more readily than those emanating from the point of confluence of tip radii.
- ii. Cracks emanating from the confluence of tip radii exhibit mixed mode conditions around the crack front and maintain a mixed mode condition as the crack front turns, exhibiting strong resistance to further crack growth.
- iii. Normalized stress intensities based upon a planar semi-elliptic shape are computed at maximum depth and near the fin surface.
- iv. While cracks emanating from the fin tip on the axis of symmetry remain essentially planar, the off-axis cracks generate curved surfaces.

The preliminary study described above suggests that the use of off-axis cracks may be an unnecessary complication in predicting crack growth and SIF determination due to the substantial irregularities and

highly elevated loads required to induce sufficient crack growth in off-axis cracks to shed the shear mode for the geometry studied. However, further experiments are underway in order to further extend the results.

### ACKNOWLEDGMENTS

The authors wish to gratefully acknowledge the support of Sparta Inc. under sub-contract No. 98522 through the Air Force Research Laboratory.

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## APPENDIX A- Frozen Stress Photoelasticity

When a transparent model is placed in a circularly polarized monochromatic light field and loaded, dark fringes will appear which are proportional to the applied load. These fringes are called stress fringes or isochromatics, and the magnitude of the maximum in-plane shear stress is a constant along a given fringe. Some transparent materials exhibit mechanical diphasic characteristics above a certain temperature, called the critical temperature ( $T_c$ ). The material, while still perfectly elastic, will exhibit a fringe sensitivity of about twenty times the value obtained at room temperature, and its modulus of elasticity will be reduced to about one six-hundredth of its room temperature value. By raising the model temperature above  $T_c$ , loading, and then cooling slowly to room temperature, the stress fringes associated with  $T_c$  will be retained when the material is returned to room temperature. Since the material is so much more sensitive to fringe generation above  $T_c$  than at room temperature, fringe recovery at room temperature upon unloading is negligible. The model may then be sliced without disturbing the "frozen in" fringe pattern and analyzed as a two-dimensional model but containing the three-dimensional effects. In the use of the method to make measurements near crack tips, due to the need to reduce loads above critical temperature to preclude large local deformations, and the use of thin slices, few stress fringes are available by standard procedures. To overcome this obstacle, a refined polariscope is employed to allow the tandem use of the Post and Tardy methods to increase the number of fringes available locally.

In fringe photographs, integral fringes are dark in a dark field and bright in a bright field. Bright fields are

used throughout this paper.

## APPENDIX B

### MODE I ALGORITHM

Beginning with the Griffith-Irwin Equations, we may write, for Mode I, for the homogeneous case,

$$\sigma_{ij} = \frac{K_1}{(2\pi r)^{\frac{1}{2}}} f_{ij}(\theta) + \sigma_{ij}^o \quad (i, j = n, z) \quad (1)$$

where:

$\sigma_{ij}$  are components of stress

$K_1$  is SIF

$r, \theta$  are measured from crack tip (Fig. B-1)

$\sigma_{ij}^o$  are non-singular stress components.

Then, along  $\theta = \pi/2$ , after truncating  $\sigma_{ij}$

$$\tau_{nz}^{\max} = \frac{K_1}{(8\pi r)^{\frac{1}{2}}} + \tau^o = \frac{K_{AP}}{(8\pi r)^{\frac{1}{2}}} \quad (2)$$

where:  $\tau^o = f(\sigma_{ij}^o)$  and is constant over the data range

$K_{AP}$  = apparent SIF

$\tau_{nz}^{\max}$  = maximum shear stress in  $nz$  plane

$$\therefore \frac{K_{AP}}{\bar{\sigma}(\pi a)^{\frac{1}{2}}} = \frac{K_1}{\bar{\sigma}(\pi a)^{\frac{1}{2}}} + \frac{\sqrt{8}\tau^o}{\bar{\sigma}} \left(\frac{r}{a}\right)^{\frac{1}{2}} \quad (3)$$

where (Fig. B-1)  $a$  = crack length, and  $\bar{\sigma}$  = remote normal stress

i.e.  $\frac{K_{AP}}{\bar{\sigma}(\pi a)^{\frac{1}{2}}}$  vs.  $\sqrt{\frac{r}{a}}$  is linear.

Since from the Stress-Optic Law

$$\tau_{nz}^{\max} = \frac{nf}{2t} \text{ where}$$

$n$  = stress fringe order

$f$  = material fringe value

$t$  = specimen (or slice) thickness

and from Eq. 2

$$K_{AP} = \tau_{nz}^{\max} (8\pi r)^{\frac{1}{2}} = \frac{nf}{2t} (8\pi r)^{\frac{1}{2}},$$

then  $K_{AP}$  (through a measure of  $n$ ) and  $r$  becomes the measured quantity from the stress fringe pattern at different points in the pattern.

In the present study, instead of normalizing  $K$  with respect to  $\bar{\sigma}(\pi a)^{1/2}$ , we have selected  $p\sqrt{\pi a/Q}$  as the normalizing factor where  $\sqrt{Q}$  is an elliptic integral of the second kind approximated here, as shown in Table I. An example of the determination of  $F_1$  in Table I from test data is given in Fig. B-2.

#### MIXED MODE ALGORITHM

The mixed mode algorithm was developed (see Fig. B-3) by requiring that:

$$\lim_{\Theta_m \rightarrow \Theta_m^0} \left\{ (8\pi r_m)^{1/2} \frac{\partial(\tau_{nz}^{\max})}{\partial \Theta} (K_1, K_2, r_m, \Theta_m, \tau_{ij}) \right\} = 0 \quad (4)$$

which leads to:

$$\left( \frac{K_2}{K_1} \right)^2 - \frac{4}{3} \left( \frac{K_2}{K_1} \right) \cot 2\Theta_m^0 - \frac{1}{3} = 0 \quad (5)$$

By measuring  $\Theta_m^0$ , which is approximately in the direction of the applied load,  $K_2/K_1$  can be determined.

Then writing the stress optic law as:

$$\tau_{nz}^{\max} = \frac{fn}{2t} = \frac{K_{AP}^*}{(8\pi r)^{\frac{1}{2}}}$$

one may plot  $\frac{K_{AP}^* \sqrt{Q}}{p\sqrt{\pi a}}$  vs.  $\sqrt{r/a}$  as before, locate a linear zone and extrapolate to  $r = 0$  to obtain  $K^*$ .

Now  $K^*$ , may be written as:

$$K^* = [(K_1 \sin \Theta_m^0 + 2K_2 \cos \Theta_m^0)^2 + (K_2 \sin \Theta_m^0)^2]^{\frac{1}{2}} \quad (6)$$

Knowing  $K^*$  and  $\Theta_m^0$ ,  $K_1$  &  $K_2$  can be determined from Eqs. 5 and 6. Details are found in Smith and Kobayashi (1993).

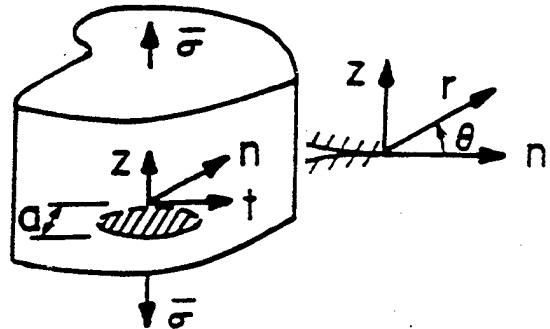


Fig. B-1 Near Tip Notation for Mode I

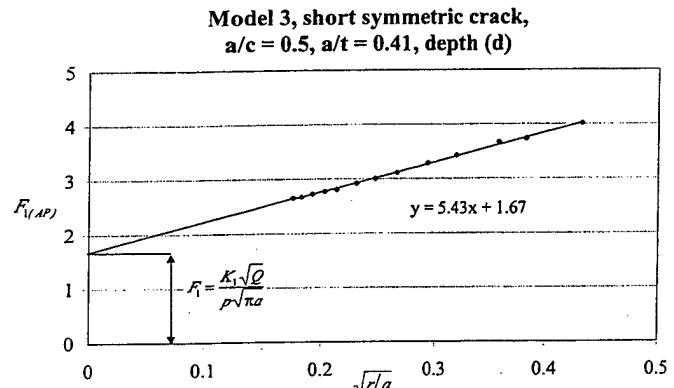


Fig. B-2 Determination of  $F_1$  from Test Data

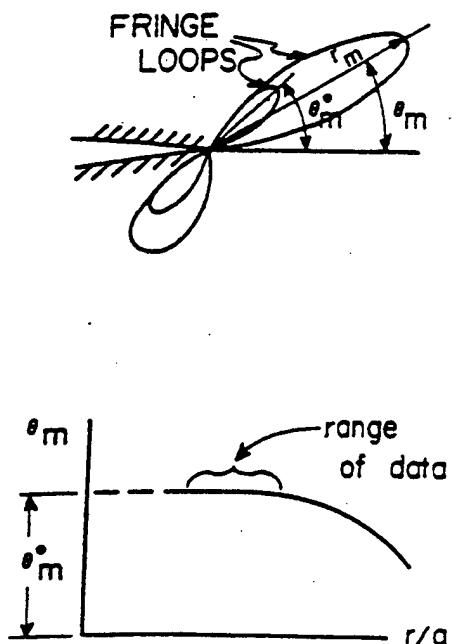


Fig. B-3 Determination of  $\theta_m^\circ$  for Mixed Mode

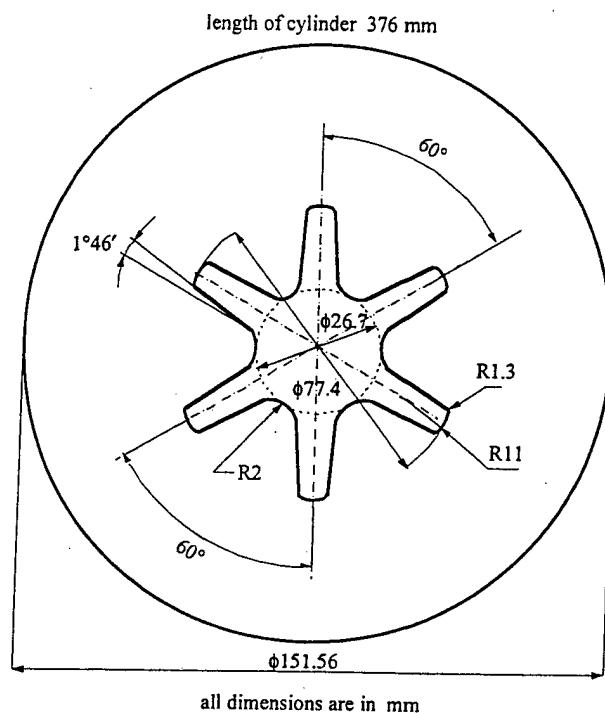


Fig. 1 Model Dimensions

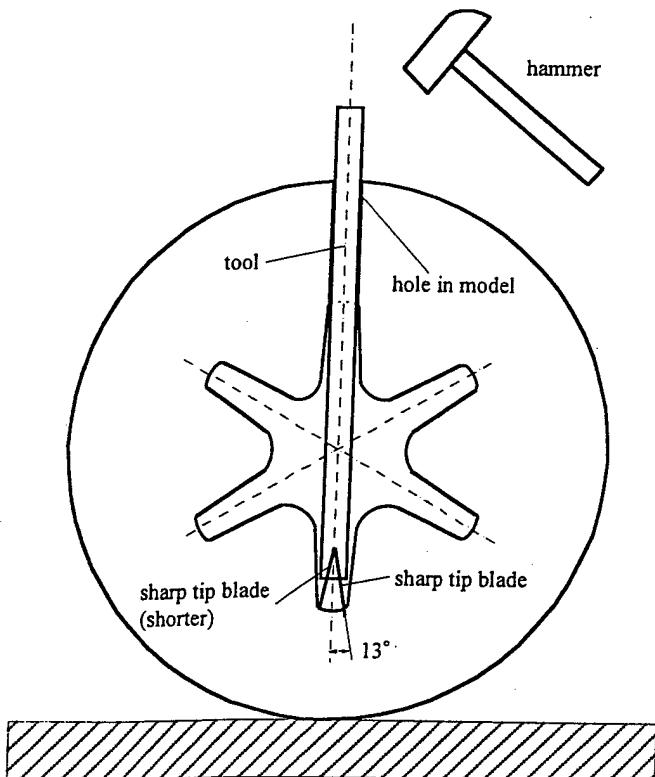


Fig. 2 Setup for Producing Off-Axis Starter Crack

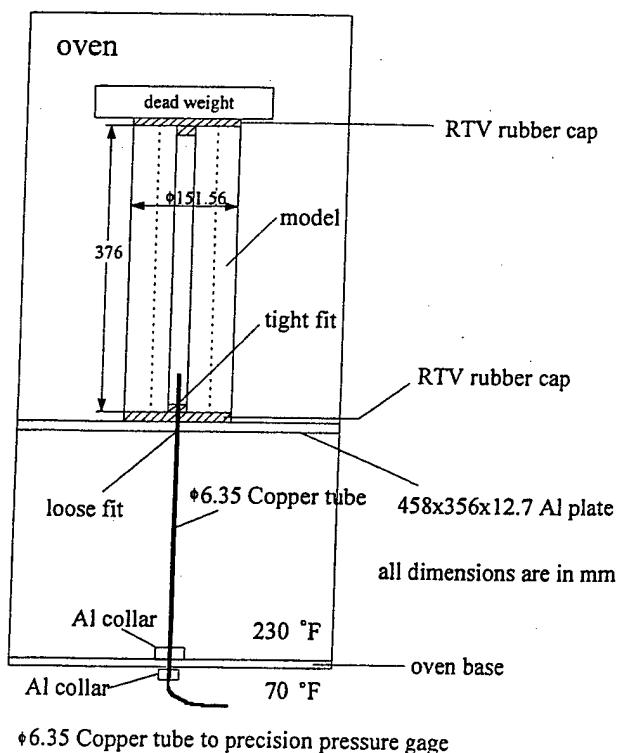


Fig. 3 Stress Freezing Test Setup

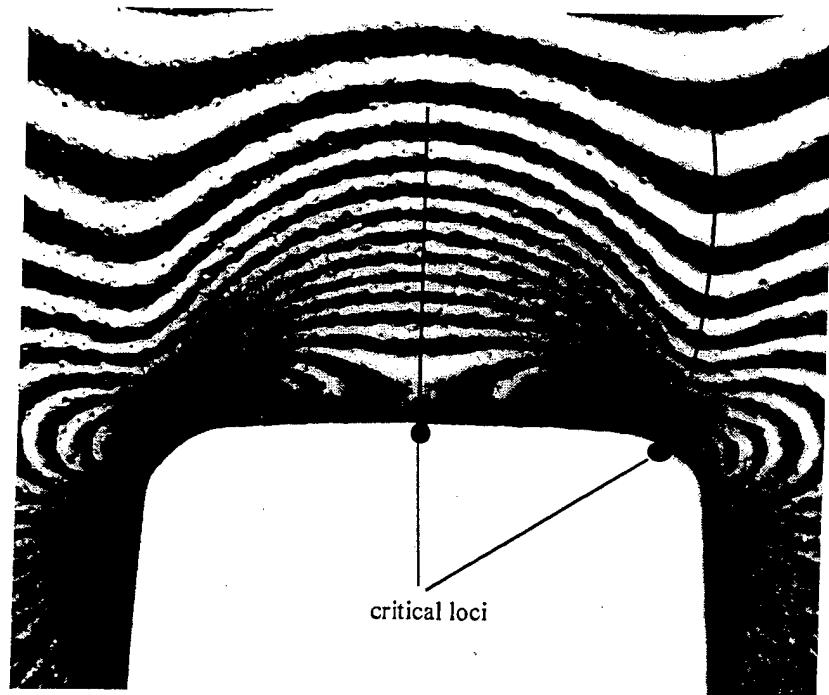


Fig. 4 Fringe Patterns Near Critical Loci at Fin Tip

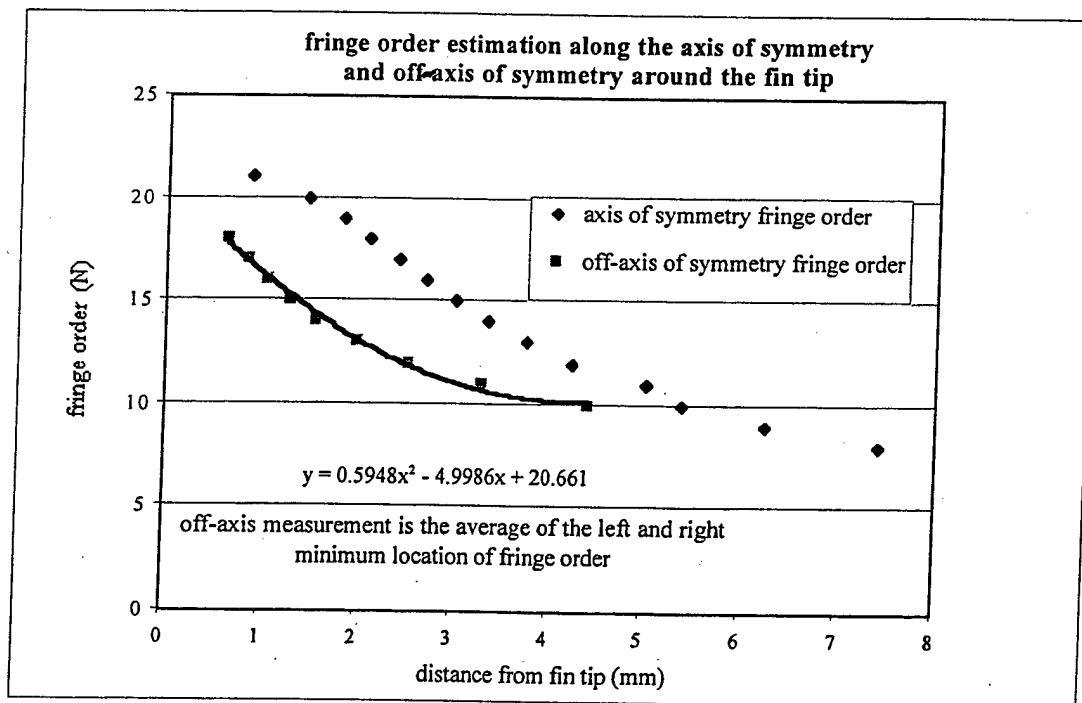
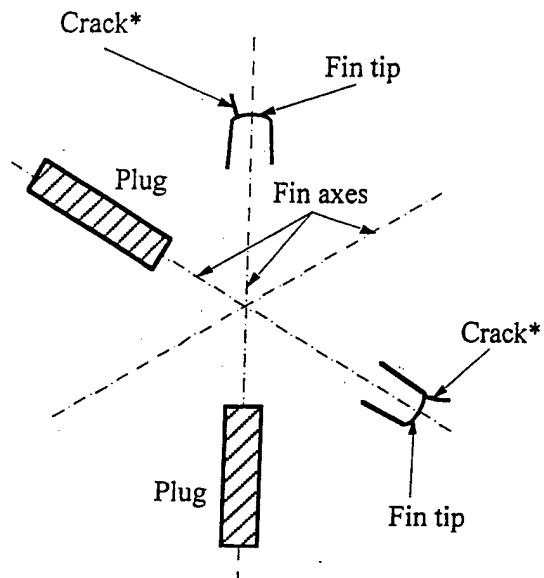


Fig. 5 Convergence of Fringe Orders at Critical Loci in Uncracked Fin



\* Path of crack to maximum depth

Fig. 6 Locations of Cracks in Model

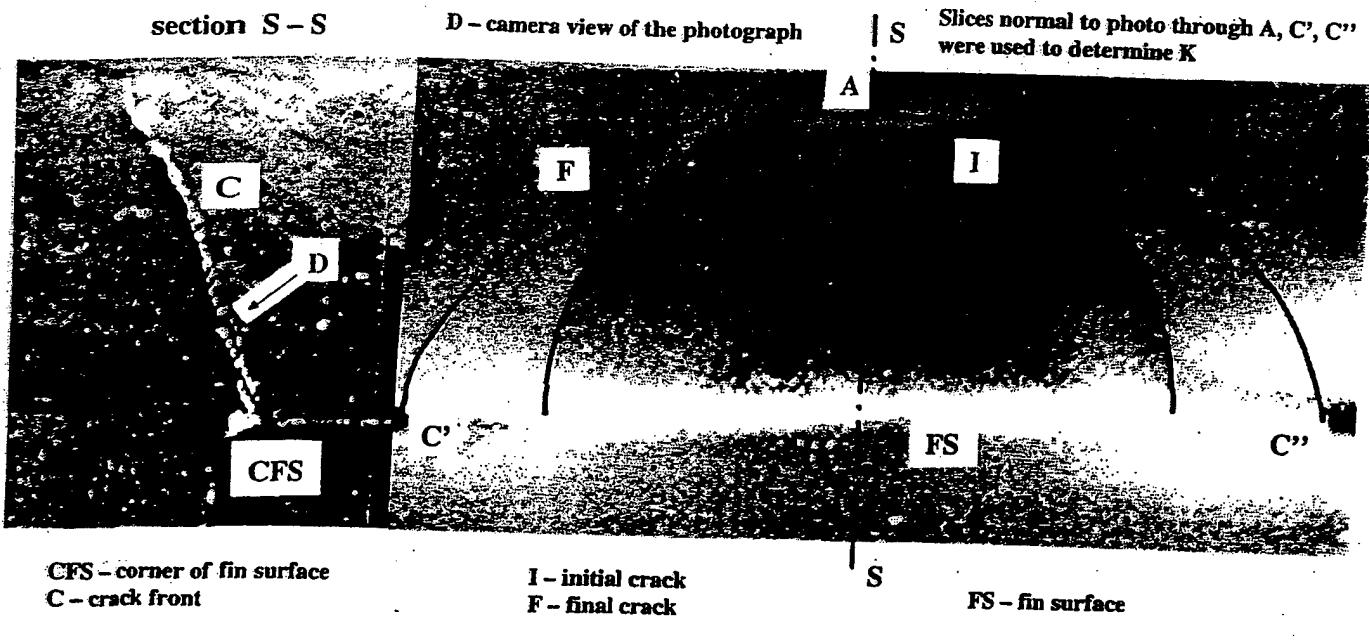


Fig. 7 A Long Off-Axis Crack (Model 2)